High-pressure structural, elastic, and electronic properties of the scintillator host material KMgF₃

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The high-pressure structural behavior of the fluoroperovskite KMgF₃ is investigated by theory and experiment. Density functional calculations were performed within the local density approximation and the generalized gradient approximation for exchange and correlation effects, as implemented within the full-potential linear muffin-tin orbital method. In situ high-pressure powder x-ray diffraction experiments were performed up to a maximum pressure of 40 GPa using synchrotron radiation. We find that the cubic $Pm\bar{3}m$ crystal symmetry persists throughout the pressure range studied. The calculated ground state properties—the equilibrium lattice constant, bulk modulus, and elastic constants—are in good agreement with experimental results. By analyzing the ratio between the bulk and shear moduli, we conclude that KMgF₃ is brittle in nature. Under ambient conditions, KMgF₃ is found to be an indirect gap insulator, with the gap increasing under pressure.

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I. INTRODUCTION

KMgF₃ is a technologically important fluoroperovskite. For example, it is used as a vacuum-ultraviolet-transparent material for lenses in optical lithography steppers and in electro-optical applications. When doped with lanthanide ions, it is a very promising material for scintillators and radiation dosimeters. In addition, the physical properties of KMgF₃ may have implications for understanding the Earth’s lower mantle.

KMgF₃ was first synthesized by van Arkel and has a simple cubic perovskite structure at room temperature. KMgF₃ demonstrates great stability under high compression and has not been found to undergo any phase transition at any temperature or pressure, suggesting it may be used as an internal x-ray calibrant.

Several experimental studies of the ground state properties of KMgF₃ have been performed. The elastic constants at ambient pressure have been measured by Rosenberg and Wigmore and by Reshchikova, while Jones investigated their pressure and temperature dependencies.

From the theoretical side, electronic structure calculations for KMgF₃ have been carried out by means of linear combination of atomic orbitals, including the effects of doping transition metal impurities in KMgF₃. The electronic structures of divalent 3d transition metal impurities doped in KMgF₃ have been investigated by the pseudopotential method, and the properties of vacancies were studied by Hartree-Fock cluster calculations. The structural, electronic, and optical properties of KMgF₃ were recently investigated by the full-potential linear augmented plane wave (FP-LAPW) method.

The present work is a combined theoretical and experimental study of the ground state and high-pressure properties of KMgF₃. We present the equation of state resulting from high-pressure diamond-anvil cell experiments on KMgF₃ up to 40 GPa. We also present the equation of state, the elastic constants, and the electronic structure from theoretical calculations using two different approximations for the exchange-correlation functional.

The remainder of the paper is organized as follows. Details of the computational method as well as details of the experimental setup are outlined in Sec. II. The measured and calculated equations of state are presented in Sec. III together with calculated ground state properties and elastic properties. The electronic structure and the pressure variation of the band gap are discussed in Sec. IV. Finally, conclusions are given in Sec. V.

II. COMPUTATIONAL AND EXPERIMENTAL DETAILS

A. Electronic structure method

The all-electron full-potential linear muffin-tin orbital (FP-LMTO) method is used to calculate the total energies and basic ground state properties of KMgF₃ presented here. In this method, the crystal volume is split into two regions: nonoverlapping muffin-tin spheres surrounding each atom, and the interstitial region between the spheres. We used a double $\kappa$ spdf LMTO basis (each radial function within the spheres is matched to a Hankel function in the interstitial region) to describe the valence bands. In the calculations, we included the 3s, 3p, 4s, 4p, and 3d bases for potassium, the 3s, 2p, 3p, and 3d bases for magnesium, and the 2s and 2p bases for fluorine. The exchange-correlation potential was calculated within the local density approximation (LDA) as well as the generalized gradient approximation (GGA) scheme. The charge density and potential inside the muffin-tin spheres were expanded in terms of spherical harmonics up to $l_{max}=6$, while in the interstitial region, they were expanded in plane waves, with 14 146 waves (energy up to 156.30 Ry) included in the calculation. Total energies...
were calculated as a function of volume for a \((16 \times 16 \times 16)\) \(k\) mesh containing 165 \(k\) points in the irreducible wedge of the Brillouin zone and were fitted to the Birch equation of state\(^{23}\) to obtain the ground state properties.

The elastic constants were obtained from the variation of the total energy under volume conserving strains, as outlined in Refs. \(^{24}\) and \(^{25}\).

**B. Experimental details**

The high-pressure x-ray diffraction measurements used a sample of polycrystalline KMgF\(_3\) prepared by the solid state reaction method from high purity constituent materials as described elsewhere in several reports.\(^{26–28}\) Diffraction patterns collected at ambient temperature and pressure showed a cubic \((Pm\overline{3}m)\) symmetry with a cell parameter \(a =4.0060(2)\) Å for KMgF\(_3\) which closely agrees with earlier reports.\(^{30,36}\) High pressures were generated by a Merrill-Bassett type diamond-anvil cell. A 185 μm sample chamber was formed in a rhenium metal gasket with a preindentation of 60 μm thickness. The powder sample was loaded in the gasket with a few ruby grains and silicone fluid as pressure transmitting medium.\(^{31}\) Diffraction experiments were performed at the 16ID-B undulator beamline of the High Pressure Collaborative Access Team (HPCAT) of the Advanced Photon Source. A monochromatic x-ray beam with a wavelength of 0.4218 Å was focused down to a size of 30 × 30 μm. Diffraction images were collected with an image plate detector for an exposure time of 10 s. The distance between the sample and the detector and the inclination angle of the image plate were calibrated using a CeO\(_2\) standard.

The two dimensional images were subsequently integrated to one dimensional diffraction patterns using the FIT2D software.\(^{32}\) The cell parameters were obtained by analyzing the diffraction patterns with the JADE software package, and the \(P-V\) data obtained were fitted with a second-order Birch-Murnaghan\(^{23}\) equation of state. The standard ruby fluorescence technique and the newly proposed ruby pressure scale of Holzapfel\(^{33}\) were used to obtain the pressures in the sample chamber.

**III. GROUND STATE AND ELASTIC PROPERTIES**

Powder x-ray diffraction patterns collected at several pressures are shown in Fig. 1. On compression, the diffraction patterns remain unchanged up to 40 GPa, except for the shifts of diffraction lines caused by the decreasing lattice constant. This implies that no structural transformations occur up to 40 GPa in KMgF\(_3\). Figure 2 shows the measured equation of state of KMgF\(_3\) and compares it with theoretical curves calculated within the LDA and GGA. A better theoretical description is obtained with the LDA, which is somewhat surprising, since usually the GGA provides an improvement over LDA. At low pressures, the LDA volume is slightly smaller than the experimental one, while the situation reverses at high pressures, i.e., altogether the LDA predicts KMgF\(_3\) to be stiffer than experimental observations. GGA, on the other hand, greatly overestimates the equilibrium volume at ambient pressure, which is the main reason for the poor agreement with experiment. If the GGA curve is scaled throughout the pressure range with the error in equilibrium volume at \(P=0\), nearly perfect agreement is found with experiment (not shown).

The lattice constant and bulk modulus measured in the present work as well as values calculated within the LDA and GGA approximations are given in Table I. Results from earlier experimental and theoretical works are quoted for comparison. The bulk modulus obtained in our experiments \(B_0=71.2(2)\) GPa with \(B'_0=4.7(3)\) compares well with other experimental results listed in Table I and also with NaMgF\(_3\) which closely agrees with earlier experimental and theoretical works.

**FIG. 1.** Powder x-ray diffraction patterns recorded at various pressures up to 32 GPa. The indexing in terms of the simple cubic structure is given.

**FIG. 2.** (Color online) Equation of state of KMgF\(_3\) in the pressure range from 0 to 40 GPa. The experimental data points are marked by circles, while theoretical results obtained with LDA and GGA are shown as full (blue) and dashed (red) curves. Volumes are given relative to the experimental equilibrium volume \(V_0 =64.288\) Å.
TABLE I. Calculated lattice constants (in Å) and bulk modulus $B_0$ (in GPa) and its pressure derivative $B'_0$ of KMgF$_3$ at the theoretical equilibrium volume compared with the experiment and other theoretical calculations. The bulk moduli have been calculated both at the experimental and theoretical volumes [$B_0(V_0^{exp})$ and $B_0(V_0^0)$, respectively].

<table>
<thead>
<tr>
<th>Lattice constant</th>
<th>$B_0(V_0^{exp})$</th>
<th>$B_0(V_0^0)$</th>
<th>$B'_0$</th>
</tr>
</thead>
<tbody>
<tr>
<td>GGA$^a$</td>
<td>4.0809</td>
<td>72.01</td>
<td>4.65</td>
</tr>
<tr>
<td>LDA$^a$</td>
<td>3.9630</td>
<td>91.47</td>
<td>4.79</td>
</tr>
<tr>
<td>LDA, LAPW$^b$</td>
<td>3.91</td>
<td>90.97</td>
<td>4.64</td>
</tr>
<tr>
<td>Experiment</td>
<td>4.0060$^c$, 3.973$^c$, 3.978±0.05,$^d$</td>
<td>71.2(2)$^a$, 70.4$^b$, 4.7(3)$^a$</td>
<td></td>
</tr>
<tr>
<td></td>
<td>3.9897$^c$, 3.993$^d$, 3.9839$^g$</td>
<td>75.1$^f$, 75.6$^f$</td>
<td></td>
</tr>
</tbody>
</table>

$^a$Present work.
$^b$Reference 19.
$^c$Reference 10.
$^d$Reference 34.
$^e$Reference 36.
$^f$Reference 35.
$^g$Reference 29.
$^h$Reference 12.
$^i$Reference 13.
$^j$Reference 14.

reported recently by Liu et al.$^{37}$ [$B=76.0(1.1)$ GPa]. The lattice constant obtained within the LDA is 1.1% lower than the experimental value, while the corresponding bulk modulus is 22% higher than the experimental value, which is the usual kind of accuracy of LDA. However, the calculated LDA lattice constant from the present work agrees quite well with the experimental work when compared to the earlier FP-LAPW(LDA) calculations, in which the reported lattice constant is 2.4% lower than the experimental value.$^{39}$ The LDA bulk modulus obtained from the present calculation agrees well with the value obtained by the FP-LAPW(LDA) method. When comparing the results obtained within GGA, the lattice constant is 1.9% higher than the experimental value, whereas our results for the bulk modulus are within the spread of the experimental data. This truly excellent agreement regarding the bulk modulus is, however, a bit fortuitous. Since the calculated equilibrium volume is overestimated with GGA (and underestimated with LDA), an error—solely depending on the error in volume—is introduced in the calculated bulk modulus. Therefore, we recalculated the bulk modulus also at the experimental volume in a manner similar to our earlier work$^{38}$ (see Table I). We find that this diminishes the discrepancies between the LDA and GGA results, as expected. In addition, the LDA bulk modulus now becomes smaller than the GGA one for KMgF$_3$, and both functionals are seen to actually overestimate the bulk modulus, by approximately 14% (LDA) and 34% (GGA).

The present experiments on KMgF$_3$ relate to recent experiments performed for NaNbF$_3$ and alloys of NaNbF$_3$ and KNaF$_3$. The crystal chemistry of Na$_{1-x}$K$_x$MgF$_3$ and NaMgF$_3$ was studied in detail at ambient and at high pressures by Zhao et al.$^{37,39}$ NaMgF$_3$ undergoes a reversible phase transition from orthorhombic ($Pbmm$) to tetragonal ($P4/mmbm$) and then to cubic structure ($Pm\bar{3}m$) upon compression. These phase transitions require either compositional changes, by increasing the K concentration to 40%, or changing temperature or pressure. The structural changes in these perovskites are due to octahedral tiltings and shortening of Mg-F bonds compared to the cubic phase. A direct transformation from orthorhombic to cubic structure in NaMgF$_3$, however, requires a very high temperature (1038 K). Moreover, the transition temperature is reported to increase with pressure. The temperature dependence of the crystal structure of KMgF$_3$ was recently investigated by neutron powder diffraction by Wood et al.$^{11}$ from 4.2 to 1223 K, and the cubic symmetry was found to be stable throughout this temperature range. The thermal expansion as well as the atomic displacement parameters obtained in their experiments show that the F ions behave less anisotropically than in NaMgF$_3$ at such high temperatures. On comparing these results with the present high-pressure diffraction experiments on KMgF$_3$, one may speculate that application of pressure alone would not be sufficient to induce structural changes in KMgF$_3$ as the cubic phase is very stable. Such a phase transformation, if any, would require either application of very high temperature or a composition change in the system to achieve changes in the order parameters. Asbrink et al.$^{40}$ have studied single crystals of the transition metal bearing perovskite KMnF$_3$, which is isostructural to KMgF$_3$, under high pressure and observed a cubic-to-tetragonal phase transition at a critical pressure of 3.1 GPa. On combining these results, phase transitions from the cubic symmetry may be expected with a combination of composition change, temperature, and pressure in KMgF$_3$. A systematic study on the octahedral tilting and order parameters with other dopant compositions and the effect of external thermodynamical variables are further required to understand the phase stability of KMgF$_3$. The elastic constants of KMgF$_3$ calculated within LDA and GGA are listed in Table II where they are also compared to experimental results as well as earlier calculations. The LDA overestimates all of the $C_{11}$, $C_{12}$, and $C_{44}$ elastic constants by between 10% and 22% compared to...
The elastic constants obtained within GGA are much closer to the experimental values than are the LDA results. For instance, both $C_{11}$ and $C_{12}$ are within the experimental spread. Of course, the elastic constants also depend sensibly on the volume, and, therefore, the same argument as for the bulk modulus can be applied here. We have, however, refrained from recalculating all the elastic constants with the volume correction, but wish to mention that the excellent agreement between experiment and the GGA elastic constants should be interpreted with care. Another point of caution is the fact that the calculated values pertain to 0 K, while experiments are performed at room temperature. Finite temperature generally tends to reduce the elastic constants because of thermal expansion. Using the calculated elastic constants, we calculated the anisotropy factor $A = 2C_{44}/(C_{11} - C_{12})$. We find an $A = 0.91$ for LDA and $A = 1.12$ for GGA. The experimental result is 1.05, measured at room temperature, which is closer to but lower than the GGA value. However, the anisotropy factor is found to decrease as the temperature is lowered.

A simple relationship, which empirically links the plastic properties of materials with their elastic moduli, was proposed by Pugh.\textsuperscript{41} The shear modulus $G$ represents the resistance to plastic deformation, while the bulk modulus $B$ represents the resistance to fracture. A high $B/G$ ratio is associated with ductility, whereas a low value corresponds to brittle nature. The critical value which separates ductile and brittle materials is around 1.75; i.e., if $B/G > 1.75$, the material behaves in a ductile manner; otherwise, the material behaves in a brittle manner. Frantsevich \textit{et al.},\textsuperscript{42} in a similar fashion, have suggested $B/G \sim 2.67$ as the critical value separating brittle and ductile behaviors. In the case of KMgF\textsubscript{3}, the calculated value of $B/G$ is 1.5 within LDA and 1.4 within GGA, hence classifying this material as brittle.

Petitfor\textsuperscript{43} suggested that the angular character of atomic bonding in metals and compounds, which also relates to the ductility, could be described by the Cauchy pressure $C_{12} - C_{44}$. For metallic bonding, the Cauchy pressure is typically positive. On the other hand, for directional bonding with angular character, the Cauchy pressure is negative, with larger negative pressure representing a more directional character. These correlations have been verified for ductile materials such as Ni and Al that have typical metallic bonding, as well as for brittle semiconductors such as Si with directional bonding.\textsuperscript{43} In the ionic compound KMgF\textsubscript{3}, the calculated Cauchy pressure is $-10$ GPa within LDA and $-15$ GPa within GGA, in good agreement with the nonmetallic characteristics of KMgF\textsubscript{3}.

Table III presents sound velocities as derived from the calculated elastic constants.\textsuperscript{24} The calculated sound velocities agree quite well with the experiments, in particular, for the GGA values, which is a consequence of the somewhat fortuitous good agreement between the measured and GGA calculated elastic constants.

### IV. ELECTRONIC STRUCTURE

The calculated electron band structure of KMgF\textsubscript{3} is shown in Fig. 3, with the ensuing density of states in Fig. 4. The valence bands consist of the F $p$ bands with a gap of 7.24 eV to the conduction band, which is dominated by K states. The LDA bands are almost identical, however, with a

![FIG. 3. Band structure of KMgF\textsubscript{3} (using GGA, at the experimental lattice constant). The zero of energy is set at the position of the conduction band minimum.](image_url)

TABLE II. Calculated elastic constants, shear modulus ($G$), and Young’s modulus ($E$) all expressed in GPa, and Poisson’s ratio $\nu$ of KMgF\textsubscript{3} at the theoretical equilibrium volume.

<table>
<thead>
<tr>
<th></th>
<th>$C_{11}$</th>
<th>$C_{12}$</th>
<th>$C_{44}$</th>
<th>$G$</th>
<th>$E$</th>
<th>$\nu$</th>
<th>Reference</th>
</tr>
</thead>
<tbody>
<tr>
<td>GGA</td>
<td>137.0</td>
<td>39.5</td>
<td>54.6</td>
<td>52.3</td>
<td>126.3</td>
<td>0.208</td>
<td>Present work</td>
</tr>
<tr>
<td>LDA</td>
<td>177.0</td>
<td>48.7</td>
<td>58.7</td>
<td>60.9</td>
<td>149.5</td>
<td>0.228</td>
<td>Present work</td>
</tr>
<tr>
<td>LDA</td>
<td>119.26</td>
<td>38.26</td>
<td>63.23</td>
<td></td>
<td></td>
<td></td>
<td>19</td>
</tr>
<tr>
<td>Experiment</td>
<td>132±1.5</td>
<td>39.6±1.5</td>
<td>48.5±0.6</td>
<td></td>
<td></td>
<td></td>
<td>12</td>
</tr>
<tr>
<td></td>
<td>138±0.2</td>
<td>43.6±0.2</td>
<td>49.8±0.08</td>
<td></td>
<td></td>
<td></td>
<td>13</td>
</tr>
<tr>
<td></td>
<td>138.5±0.5</td>
<td>44.1±0.5</td>
<td>50.01±0.1</td>
<td></td>
<td></td>
<td></td>
<td>14</td>
</tr>
</tbody>
</table>

**TABLE III.** Calculated longitudinal, shear, and average wave velocities ($v_l$, $v_s$, and $v_m$, respectively) in m/s for KMgF\textsubscript{3} at the theoretical equilibrium volume.

<table>
<thead>
<tr>
<th></th>
<th>$v_l$</th>
<th>$v_s$</th>
<th>$v_m$</th>
</tr>
</thead>
<tbody>
<tr>
<td>Present work</td>
<td>LDA</td>
<td>7402</td>
<td>4396</td>
</tr>
<tr>
<td>GGA</td>
<td>6706</td>
<td>4073</td>
<td>4507</td>
</tr>
<tr>
<td>Experiment</td>
<td>6470\textsuperscript{a} 6540\textsuperscript{b}</td>
<td>3940\textsuperscript{a} 3900\textsuperscript{b}</td>
<td>4290\textsuperscript{c}</td>
</tr>
</tbody>
</table>

\textsuperscript{a}Wave vector along $<100>$ direction (Ref. 12).
\textsuperscript{b}Wave vector along $<110>$ direction (Ref. 12).
\textsuperscript{c}Reference 11.
the valence band maximum occurs at the \( \Gamma \) point, while the conduction band minimum occurs at the \( R \) point (1/2, 1/2, 1/2) \( \pm a \). The largest occupied energy level at the \( M \) point (1/2, 1/2, 0) \( \pm a \) is marginally lower (by \( \sim 0.02 \text{ eV} \)) than the valence band maximum at \( R \), and it remains lower throughout the pressure range studied here.

### V. CONCLUSIONS

In conclusion, we find that pure cubic KMgF\(_3\) is very stable under high compression. From our analysis, we also find that it is a brittle system and an indirect gap insulator whose gap increases with pressure.

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